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Enhancing Calendering Process Conditions by Blending Poly(vinyl chloride) with Polyethylene, Polypropylene & Poly(methyl methacrylate)

OZAN TOPRAKCI^{1*}, ABDULLAH TAYYIB AKCAY² and HATICE AYLIN KARAHAN TOPRAKCI³

^{1, 3}Department of Polymer Materials Engineering, Yalova, Turkey. ²Institute of Graduate Studies, Yalova University, Yalova, Turkey.

Abstract

In this study, blend films were prepared using the calendering process, with polyethylene (PE), polypropylene (PP), and poly(methyl methacrylate) (PMMA) melt-mixed with poly(vinyl chloride) (PVC) at various ratios. The process conditions, structural, thermal, morphological, and mechanical properties were analyzed using Fourier transform infrared spectroscopy, differential scanning calorimetry, scanning electron microscopy, and mechanical testing. Results indicated that PVC/PE and PVC/PP blends were immiscible, while PVC/PMMA blends were miscible. The miscibility directly enhanced the mechanical properties, with yield stress and elastic modulus of PVC/PMMA (95/5 wt%) blends improving by 29.7% and 28.8%, respectively. Calendering conditions, such as banking time and current values, improved by 2-32% for all PVC blends. The increased miscibility significantly reduced energy consumption and production time, offering a cost-effective method for calendering PVC films.

Article History

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Keywords

Calendering; Polymer Blend Films; Miscibility; PVC/PMMA Blends.

Introduction

Calendering is one of the oldest techniques in polymer processing, used for fabricating highvolume plastic films or sheets.^{1,2} It is a simple twostep process involving the melting of the polymer and film formation by calendering rolls. Molten polymer passes through a set of rollers to create a uniform thickness and smooth surface. This process is crucial for polymer processing as it involves compressing and shaping the polymer material into a uniform thickness, enhancing its mechanical and surface properties. It ensures precise control over the thickness and dimensions of the final product, enabling the production of high-quality films, sheets, and other flat polymeric products used in various industries such as packaging, construction, automotive, textile, electrical, and medical sectors.^{2,3}

CONTACT Ozan Toprakci 🔀 ozan.toprakci@gmail.com 🖓 Department of Polymer Materials Engineering, Yalova, Turkey.



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Poly(vinyl chloride) (PVC) is a widely used polymer in the calendering process.^{4–7} Due to the versatility, durability, and cost-effectiveness of PVC films, they are utilized in numerous industrial applications including packaging (shrink-wraps, cling films, blister packs), construction (wall coverings, flooring, window films, banners), automotive (interior trims, protective films), medical (intravenous bags, tubes), and electrical (insulative or protective films).8 However, neat PVC cannot be melt-processed due to its low thermal degradation temperature (approximately 80°C).9 Therefore, PVC must be mixed with suitable materials such as plasticizers,10 heat stabilizers,¹¹ internal or external lubricants,¹² and impact modifiers,13 to improve its processability and the final product properties.

Despite these enhancements, plasticizer migration in PVC films can lead to a decrease in mechanical properties and compromise the film's long-term performance.¹⁴ Additionally, surface imperfections like air entrapment and defects during the calendering process can affect the visual appearance and functionality of the final product. Another significant drawback is the high energy consumption during polymer film processing by calendaring, even though it is cheaper than polymer casting route.⁷ Thus, new approaches to reduce polymer processing costs are of great importance, such as multi-layer calendering,¹⁵ innovative roller designs,¹⁶ new roll-heating routes,¹⁷ or process optimization using additives.¹⁸

While organic and inorganic-based materials can be used as additives, low molecular weight and/or incompatible additives generally worsen the mechanical properties of the films. Miscible polymer blends present a promising solution for enhancing the aesthetic and mechanical properties of films while also reducing production costs in the calendering process. Further study is required to enable convenient polymer blending strategies to be used in practical calendering applications. The novelty of this study lies in exploring the potential of polymer blends to address these challenges effectively. By focusing on the blend of PVC with polymers like polymethyl methacrylate (PMMA),19 polypropylene (PP),²⁰ and polyethylene (PE),²¹ this research aims to offer innovative solutions that improve both mechanical properties and processing efficiency in the calendering process.

In this study, PVC/polymer blend films were prepared using the calendering process. Various amounts of PMMA, PP, and PE polymers were melt-mixed with plasticized PVC to obtain polymer blend films. Structural, morphological, thermal, and mechanical behavior of polymer blend films were characterized using various techniques. Additionally, the effects of miscibility on processing conditions were investigated. The results showed that polymer blending has significant potential in the calendering process of PVC films, offering improvements in both mechanical properties and processing efficiency.

Materials and Methods Materials

CGCP, Thailand provided suspension-grade PVC with a K value of 65–67. In this work the composition of PVC compound was directly taken from an industrial available recipe from the production line. PP, PE, PMMA and reference PVC that consisted of PVC (100 phr), plasticizer (3 phr), CaCO₃ (12.5 phr), TiO₂ (7.5 phr), thermal stabilizer (2.75 phr), internal lubricant (1 phr), external lubricant (0.5 phr) and impact modifier (5 phr) were supplied by a local supplier.

Preparation of Polymer Blend Films

As given in Figure 1, polymer blend films were produced in three steps. In the first step, the reference PVC compound was prepared by using a hot/cold mixer (Dersan, DM-LB10) at ~110°C, 750 rpm for ~8 min and ~45°C, 250 rpm for ~3 min, respectively. In the second step, blending of PVC with PE, PP and PMMA was performed at different ratios (wt.%) by using a laboratory scale blender (Entex, L-WE30) at ~165°C and 100 rpm for ~30 min. In the final step, PVC compound and PVC/polymer blends were calendered by using labscale calendering machine with two rollers. During calendering process, banking times and electrical current values were also measured. PVC blend compositions and their properties were given in Table 1. In addition to compositional information, Table 1 gives electrical current values during the process. It is obvious from the results, all blends resulted in lower current values that is an indication of lower energy consumption and decrease in processing costs. Furthermore, banking times were decreased in the range of 2.5 - 12% depending on blend composition. So, blending PVC with other polymers can improve production speeds and

synergistically decrease processing cost. It should be noted that 90PVC/10PP and 90PVC/10PE blend films could not be fabricated by calendering.

Characterization of Polymer Blend Films

The structural behavior of polymer blend films was characterized by using attenuated total internal reflectance (ATR) mode (Perkin Elmer, Spectrum-2 IR spectrometer) of a Fourier transform infra-red spectroscopy (FTIR) system. Between 450 and 4000 cm⁻¹, spectra were captured in the transmittance mode at a scan rate of 4 scans, with a spectral resolution of 4 cm⁻¹.

The thermal behavior of polymer blend films was examined by differential scanning calorimetry (DSC, Seiko). Under N₂ atmosphere, one cooling and two heating scans were conducted between -70°C and 180°C, at 5°C min⁻¹ cooling/heating rate, with an amplitude of 1°C, and a time of 60 s. The total run time per sample was ~2 h and the sample weighed ~7-8 mg. The glass transition (T_g) temperature was calculated from the reversing heat flow curves using the second heating scan in order to produce a comparable thermal history.



Fig. 1: Schematic representation of polymer blend film production route by calendering process

Before calendaring, PVC compound is produced with hot/cold mixing. Then, PVC compound is blended with suitable polymer in blender, and finally, calendaring process is done to produce PVC blend film.

To characterize the morphology of polymer blend films, JEOL JSM-6400, 30kV field emission scanning electron microscopy (FESEM) was utilized. A blade was employed to cut the polymer mix films, and Au/ Pd alloy was sputter coated onto the cross-section of the sample prior to analysis with a coating thickness in the range of 3 – 6 nm.

Utilizing universal testing equipment (Devotrans, DVT GPU/RD), the mechanical properties of polymer blend films, such as elastic modulus, yield point, their tensile strength and tensile strain, were assessed.

A blade was used to cut the samples with certain dimensions (50 mm x 5 mm). A thickness meter (Asimeto) was used to determine the average thickness of the composites before the mechanical test. For every sample, three mechanical tests were made at a speed of 1.5 cm min^{-1} .

Results and Discussion Structural Analysis

FTIR spectroscopy was carried out to analyze the structural behavior of PVC polymer blend films. FTIR spectra were recorded in transmittance mode in between 4000 – 450 cm⁻¹ region and given in Figure 2. The chemical structures of polymers used in composite films were also shown in Figure 2 a-c. The characteristic peaks belonging to IR spectrum of PVC, PMMA, PE and PP can also be seen from Table 2. The characteristic peaks of plasticized PVC

films can be seen in Fig.2. The peaks at 2851, 2922, and 2958 cm⁻¹ were assigned to -CH stretching vibrations. CH₂-Cl angular deformation vibration at 1425 cm⁻¹, CH-Cl out of plane angular deformation vibration at 1244 cm⁻¹ and C-Cl bond stretching vibrations at 874 cm⁻¹ were observed for all samples.

The peaks at 964 and 608 cm⁻¹ represented trans C-H wagging, and cis C-H wagging vibrations, respectively. The peaks at 1735 and 1330 cm⁻¹ were related to -C=O stretching vibrations of carbonyl group of plasticizer and deformation vibrations of -CH₂.²²⁻²⁴

Sample ID#	Polymer Blends	Ratio (wt%)	Film Formation	Surface Properties	Current Values During Calendering (A)	Banking Time (s)
PVC 99PVC/1PP 95PVC/5PP 90PVC/10PP 99PVC/1PE 95PVC/5PE 90PVC/10PE 99PVC/1PMMA 95PVC/5PMMA 90PVC/10PMMA	PVC PVC/PP PVC/PP PVC/PE PVC/PE PVC/PE PVC/PE PVC/PMMA PVC/PMMA	100 99/1 95/5 90/10 99/1 95/5 90/10 99/1 95/5 90/10	>>>×>>>>>>>>>>>>>>>>>>>>>>>>>>>>>>>>>>	Smooth Rough X Rough Rough X Smooth Smooth Smooth	37 32 (-13.5%↓) 28 (-24.3%↓) × 30 (-18.9%↓) 25 (-32.4%↓) × 34 (-8.1%↓) 30 (-18.9%↓) 32 (-13.5%↓)	85 83 (-2.35%↓) 77 (-9.41%↓) × 78 (-8.24%↓) 75 (-11.76%↓) × 82 (-3.53%↓) 80 (-5.88%↓) 81 (-4.71%↓)

Table 1: PVC blend compositions and their properties

For PVC/PE films, when PE content in the blend increased to 5%, the peaks at 2851 and 2922 cm⁻¹ become more distinct, these peaks were assigned to symmetrical and asymmetrical C-H stretching vibrations of PE, respectively. Also, CH_2 -rocking and scissoring vibration peaks at 711 and 1459 cm⁻¹

could be seen at Fig. 2a.²⁵ For PVC/PP blend films, the peaks at 1450, 2851, 2918 cm⁻¹ were assigned to -CH₂- symmetrical bending, -CH₂- symmetrical stretching and -CH₂- asymmetrical stretching, respectively.²⁶ Also, symmetrical bending vibration of -CH₂ group was detected at 1376 cm⁻¹ (Fig.2b).²⁷



Fig. 2: FTIR spectra of different polymer blend films. a) PVC/PE blends, b) PVC/PP blends, c) PVC/ PMMA blends and d) comparison of all PVC blends with 5% additive polymer. Colored bands show important wavenumbers of functional groups

For PVC/PMMA blend films, the peaks at 2918 and 1731 cm⁻¹ represented stretching vibrations of C-H, and C=O groups, respectively (Fig.2c). The peaks at 875, 1192, 1427, 1731, and 2949 cm⁻¹ were related to stretching vibrations of C-CI, -O-CH₃, CH₃, C=O, and C-H groups in the polymer blends, respectively. In parallel with the literature, the characteristic peaks of plasticized PVC 1735, 1244, 670 cm⁻¹ were shifted

to 1731, 1247 and 682 cm⁻¹ in PVC/PMMA polymer blend films.^{23,28-30} Also, intensities of the carbonyl peaks were increased with the increment of PMMA amount in PVC/PMMA polymer blends. As previously described by Soman and Kelkar, miscibility might be enhanced by a particular hydrogen bonding interaction between the carbonyl group (C=O) of PMMA and the hydrogen from (-HC-CI) of PVC.²²

Vibrational	Wavenumber (cm ⁻¹)					
Modes	PVC	95PVC/5PMMA	95PVC/5PP	95PVC/5PE		
C-H stretching	2851, 2922, 2958	2850, 2918, 2949	2851, 2918, 2958	2851, 2922, 2958		
C=O stretching	1735	1731	1734	1735		
CH, deformation	1425, 1459	1427	1425,1450	1425,1459		
-CH ₃ bending			1376			
CH, bending	1330	1330	1330	1330		
CH-Cl rocking	1244	1247	1253	1253		
-OCH ₃ stretching		1192, 1150				
-CH ₃ rocking			1167, 968			
trans CH wagging	964	964	964	964		
C-CI stretching	874, 670	875, 682	874, 678	874, 670		
-CH ₂ rocking			844 - 810	731, 711		
cis CH wagging	608	609	611	608		

Table 2: Vibational modes observed in different pol	ymer blends
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Thermal Analysis

The thermal behavior of PVC, 5PE/95PVC, 5PP/95PVC and 5PMMA/95PVC blend films as

a function of temperature is presented in Fig. 3. To make it easier to compare the transitions of the various samples, curves in the figure were intentionally moved vertically along the heat flow axis (Figure 3). The discussion was based on the second heating of every sample because the first heating and cooling was done to create a similar thermal history to all samples.

Commercial PVC is an amorphous polymer and only glass transition temperature (T_g), can be observed which can be around 70-90°C depending on the composition of PVC for film making. While neat PVC shows a a T_g of 90°C, this value decreases with the addition of plasticizer. PVC, 5PP/95PVC and 5PE/95PVC film samples showed a T_g of 72.3°C, this shows that PP or PE have no effect on T_g of PVC. It can be clearly seen that T_g value increased from 72.3°C to 75.6°C with the addition of PMMA into PVC composition (Fig. 3 inset). This is parallel with the literature.³¹ Change in the glass transition is the obvious proof the miscibility and it also shows

that PVC/PMMA polymer blends are compatible to each other.³⁰ From Fig. 3, it can also be seen cold crystallization exotherm both 5PP/95PVC and 5PE/95PVC film samples. The cold crystallization temperatures (T_{cc}) were measured as 141.02°C and 129.35°C from the maximum value of cold crystallization for 5PP/95PVC and 5PE/95PVC film samples, respectively.



Fig. 4: SEM images of different polymer blend films @ 5kx, 20kx and 50kx a) PVC film, b) 95PVC/5PE blend film, c) 95PVC/5PP blend film, and d) 95PVC/5PMMA blend film. White arrows show other polymer phases (such as PE or PP) in the polymer blend. Dashed curvy cornered rectangles show focus area in the images

Morphological Analysis

The morphology of blends is evaluated by some aspects such as scale, geometry, and spatial distribution of phase separation and so on. It is directly influenced by the type of polymers and ratio of the components. Polymer blends can be classified as miscible blends and immiscible blends due to separation of the phases. While miscible blends behave as one single phase, immiscible blends tend to exhibit continuous and dispersed phases and dispersed phase is generally observed as spherical, particle, plate or channel-like geometry in the continuous phase especially at low concentrations. In our case, PVC, 95PVC/5PP, 95PVC/5PE, 95PVC/5PMMA were chosen for morphology analysis. To analyze the phase morphology and phase interphase, FESEM images were given at different magnifications such as 5k, 20k and 50kx. While low magnification images were given to observe the continuous and dispersed phases, high magnification images are useful to observe the size, geometry and interphase of the phases. Fig. 4a is the cross-sectional image of PVC film. PVC has been successfully melted. As previously mentioned, an industrial recipe was used for preparation of PVC compound and the particles throughout the film cross-section were probably inorganic materials such as CaCO₃ and TiO₂. These particles were also

observed for all blends. 95PVC/5PP based blend film can be seen from Fig. 4b. PVC was the continuous phase and PP phase was not found to be miscible in PVC matrix. PP phase was indicated by arrows in SEM images. The morphology of PP phase was almost like stretched plates and stretched channels. 95PVC/5PE based blend film can be seen from Fig. 4c. PVC was the continuous phase and PE phase was not found to be miscible in PVC matrix that was also reported in previous studies.32 As shown by the arrows in FESEM images, the morphology of PP phase was almost like elliptical particles. 95PVC/5PE based blend film was shown Fig. 4d. Unlike PP and PE blends PMMA was found to show higher miscibility in PVC. As previously mentioned in FTIR analysis, the miscibility of PMMA in PVC was reported to be caused by interaction between -CHCI and C=O groups of PVC of PMMA, respectively.30



Fig. 5: Mechanical behavior of different polymer blends. a) PVC/PE blends, b) PVC/PP blends, c) PVC/PMMA blends and d) comparison of all PVC blends with 5% additive polymer

Sample ID#	Yield Stress (MPa)	Strain at Break (%)	Elastic Modulus (MPa)	
PVC	70.36	13.18	660	
99PVC/1PP	81.64 (16%↑)	8.20 (-37.8%↓)	740 (12.1%↑)	
95PVC/5PP	79.47 (13.9%↑)	7.85 (-40.4%↓)	742 (12.4%↑)	
99PVC/1PE	73.79 (4.9%↑)	12.77 (-3.1%↓)	695 (5.3%↑)	
95PVC/5PE	68.94 (-2.0%↓)	13.44 (1.9%↑)	654 (-0.9%↓)	
99PVC/1PMMA	87.62 (24.5%↑)	12.55 (-4.8%↓)	819 (24.0%↑)	
95PVC/5PMMA	91.26 (29.7%↑)	8.54 (-35.2%↓)	850 (28.8%↑)	
90PVC/10PMMA	96.26 (36.8%↑)	8.67 (-34.2%↓)	890 (34.8%↑)	

Table 3: PVC blend compositions and their properties

Mechanical Characterization

Stress-strain curves of PVC, PVC/PE, PVC/PP and PVC/PMMA blend films can be seen from Fig. 5. Yield stress, strain at break, elastic modulus values of the samples and rational changes compared to PVC were summarized in Table 3. Since 90PVC/10PP and 90PVC/10PE blend films could not be fabricated by calendering, their mechanical testing was not performed. PVC showed classical response with the yield strength, strain at break and elastic modulus values of 70.36 MPa, 13.18 % and 660 MPa respectively. PVC/PP composites showed increase in yield stress and elastic modulus as PVC<95PVC/5PP<99PVC/1PP. On the other hand, due to higher modulus, strain values at break showed drop as 95PVC/5PP<99PVC/1PP<PVC. PVC/PE blend showed slight increase in mechanical performance for 99PVC/1PE and decrease for 95PVC/5PE. That can be given as 95PVC/5PE<PVC<99PVC/1PE and assumed to be caused by immiscibility of PE in PVC. PVC/ PE blends showed relatively lower mechanical performance compared to PVC/PP samples as given in Table 3. That can be explained by FESEM images. As can be seen from low and high magnification images PE showed higher phase separation. By incorporation of PMMA into PVC, both yield stress and elastic modulus increased as PVC<99PVC/1PE<95PVC/5PE and strain at break decreased due to higher elastic modulus and rigid structure of PMMA. Fig. 5d shows the performance of PVC and blend films. As obvious from the graph, PMMA blends showed the highest performance and mechanical performance of the samples can be given as PVC/PE<PVC<PVC/PP<PVC/PMMA. That is also parallel with morphological analysis. As mentioned previously, PMMA was miscible in PVC due to interactions between two polymers.

Conclusions

In this study, PVC/PE, PVC/PP, and PVC/PMMA blend films were prepared using a straightforward 3-step process (melt mixing, blending, calendering). The aim was to examine the impact of blending on calendering conditions and film properties. The characterization of structural, morphological, thermal, and mechanical behavior revealed that while PVC/PE and PVC/PP blends were immiscible, PVC/PMMA blends were consistently miscible, irrespective of the PMMA ratio. This miscibility significantly enhanced the mechanical properties, with yield stress and elastic modulus of 95PVC/5PMMA blend films increasing by 29.7% and 28.8%, respectively, compared to pure PVC films. Additionally, blending with PMMA reduced energy consumption and processing costs. These findings underscore the significant potential of PVC/PMMA blends in improving the efficiency and effectiveness of the calendering process for PVC films.

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Conflict of Interest

There is no conflict of interest.

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